# Crystallography basics





<sup>a</sup>The 11 Laue symmetries are separated by horizontal lines.

# Family of planes

**(hkl)** - Family of plane: parallel planes and equally spaced. The indices correspond to the plane closer to the origin which intersects the cell at  $a/h$ , b/k and c/l.

Miller indices describe the orientation and spacing of a family of planes.

The spacing between adjacent planes of a family is referred to as the "**dspacing**".



1.ZUUL.

13 U U J

UU !

# Planes (and directions) of a form

**{hkl}** - Planes of a form: equivalent lattice planes related by symmetry. 

For the cubic system all the planes  $(100)$ ,  $(010)$ ,  $(001)$ ,  $(\bar{1}00)$ ,  $(010)$  and  $(001)$  belong to the form  $\{100\}$ . In the cubic system all the planes (100), (010), (001), (1

For a tetragonal material a=b≠c the form  $\{100\}$  would only include (100), (010), ( $\bar{1}00$ ), and (0 $\bar{1}0$ ).  $\overline{1}$  nar a-v+c tion<br> $\overline{1}$  nn  $\overline{1}$  nn  $\overline{1}$ 

**<uvw>** - Directions of a form: equivalent lattice directions related by symmetry

# Planes of a zone



### **Planes of a zone** - The planes of a zone axis **[uvw]** satisfy the Weiss Zone Law:

### $hu + kv + lw = 0$

This law is valid for all lattices, Cartesian, or not.

In cubic systems [hkl] is normal to the set of planes (hkl) and the Weiss zone law can be expressed as the scalar (dot) product of [uvw] and the plane normal [hkl].



The shaded planes in the cubic lattice are planes of the zone [001].

The planes of zone are not all of the same form. 

Any direction is a zone axis!

# Interplanar distances (d) formulae

In the case of orthogonal systems determination of interplanar distances is simple.



Intercepts of a lattice plane (*hkl) on the unit cell vectors a, b, c. As there is another plane of the same family passing through O the interplanar distance is just: ON=d<sub>hkl</sub>* 

Hence:  $(h/a)^2$ .d<sub>hkl</sub><sup>2</sup>+  $(k/b)^2$ .d<sub>hkl</sub><sup>2</sup> +  $(l/c)^2$ .d<sub>hkl</sub><sup>2</sup> = 1

As a result: 
$$
(h/a)^2 + (k/b)^2 + (l/c)^2 = 1/d_{hkl}^2
$$

# Interplanar distances (d) formulae

Cubic:

\n
$$
\frac{1}{d^2} = \frac{h^2 + k^2 + l^2}{a^2}
$$
\nTetragonal:

\n
$$
\frac{1}{d^2} = \frac{h^2 + k^2}{a^2} + \frac{l^2}{c^2}
$$
\nHexagonal:

\n
$$
\frac{1}{d^2} = \frac{4}{3} \left( \frac{h^2 + hk + k^2}{a^2} \right) + \frac{l^2}{c^2}
$$

 $Rhombohedral$ :

$$
\frac{1}{d^2} = \frac{(h^2 + k^2 + l^2)\sin^2\alpha + 2(hk + kl + hl)(\cos^2\alpha - \cos\alpha)}{a^2(1 - 3\cos^2\alpha + 2\cos^3\alpha)}
$$
  
Orthorhombic:  

$$
\frac{1}{d^2} = \frac{h^2}{a^2} + \frac{k^2}{b^2} + \frac{l^2}{c^2}
$$
  
Monoclinic:  

$$
\frac{1}{d^2} = \frac{1}{\sin^2\beta} \left(\frac{h^2}{a^2} + \frac{k^2\sin^2\beta}{b^2} + \frac{l^2}{c^2} - \frac{2hl\cos\beta}{ac}\right)
$$
  
Triclinic:  

$$
\frac{1}{d^2} = \frac{1}{V^2} (S_{11}h^2 + S_{22}k^2 + S_{33}l^2 + 2S_{12}hk + 2S_{23}kl + 2S_{13}hl)
$$

In the equation for triclinic crystals,

 $S_{33} = a^2b^2 \sin^2 \gamma$ ,

 $V =$  volume of unit cell (see below),  $S_{12} = abc^2(\cos \alpha \cos \beta - \cos \gamma),$  $S_{11} = b^2 c^2 \sin^2 \alpha$ ,  $S_{23} = a^2bc(\cos \beta \cos \gamma - \cos \alpha),$  $S_{22} = a^2 c^2 \sin^2 \beta$ ,  $S_{13} = ab^2c(\cos \gamma \cos \alpha - \cos \beta).$ 

 $\overline{7}$ 

# Symmetry operations

- A symmetry element (or operator) when applied to an object leaves that object unchanged
- An object has translational symmetry if it looks the same after a particular translation operation (an example is wallpaper, which has a repeating pattern; if you slide it by the right amount it looks the same as before).
- A point symmetry operation is specified with respect to a point in space which does not move during the operation (eg. inversion, rotation, reflection, improper rotation)

# Translational symmetry operations



**Lattice** - Infinite array of points in space, in which each point has identical surroundings. 

The simplest way to generate such na array is by using translation invariance (tranlational symmetry operation).

# Unit cell

• The repeat 3d unit in a lattice is called a unit cell



- A unit cell is defined by six parameters,  $\mathbf{a}$ ,  $\mathbf{b}$ ,  $\mathbf{c}$ ,  $\alpha$ ,  $\beta$ , and  $\gamma$ ٠
- $a \times b \cdot c = c \times a \cdot b = b \times c \cdot a =$  volume of the unit cell = V

# Unit cell choice

- There is always more than one possible choice of unit cell
- By convention the unit cell is usually chosen so that it is as small as possible while reflecting the full symmetry of the lattice
- If the unit cell contains only one lattice point is said to be primitive
- If it contains more than one lattice point it is centered

Face centered cubic **Primitive** 





Body centred cubic **Primitive** 

# Why does crystallography need symmetry?

Crystal structure of calcite, a form of calcium carbonate

The symmetry of a crystal can be used to reduce the number of unique atom positions we have to specify



# Point symmetry operations



Symmetry elements:

- **(a) Mirror plane,** shown as dashed line, in elevation and plan.
- **(b) Twofold axis,** lying along broken line in elevation, passing perpendicularly through clasped hands in plan.
- **(c) Combination of twofold axis with mirror planes**, the position of the symmetry elements given only in plan.
- **(d) Threefold axis**, shown in plan only.
- **(e) Centre of symmetry** (in centre of clasped hands)
- **(f) Fourfold inversion axis**, in elevation and plan, running along the dashed line and through the centre of the clasped hands (compound point **symmetry operation)**

# (Compound point symmetry operations)

**Compound operations:** Combinations of a rotation with a reflection or inversion. Inversion takes a locus on points.

Simple rotations are **proper;** that is, they generate a sequence of objects with the same handedness**. Improper** rotations **(roto-inversions)** produce objects of **alternating handedness.** 

**Roto-inversions** involve rotation and inversion. The overbar is used to designate rotoinversion. The figure below shows the operation of a 3-fold roto-inversion axis.



# Point symmetry operations



In written text mirror planes are given the symbol *m, while axes and the corresponding inversion axes are referred to as*  $1, \overline{1}$ ;  $2, \overline{2}$ ( $\equiv m$ ); 3,  $\overline{3}$ ; 4,  $\overline{4}$ ; 6,  $\overline{6}$ . The symbol 1 (for a onefold *axis) means no symmetry at all, while the corresponding inversion axis (* $\bar{1}$ *) is equivalent, as already remarked, to a centre of symmetry.*

Symmetry elements using conventional symbols. The righthand group of (a) is drawn here in a different orientation, and the left-hand groups of (c) and (f) are omitted. Symbols + and represent equal distances above and below the plane of the paper: open circles represent asymmetric units of one hand, and circles with commas their enantiomorphs. (a) Mirror plane (m), perpendicular to (left) and in the plane of the paper. (b) Twofold axis (2) in the plane of the paper (left) and perpendicular to it (right). (c) Combination of twofold axes and mirror planes. Note that the presence of any two of these elements creates the third. (d) Three fold axis (3). (e) Centre of symmetry (1). (f) Fourfold inversion axis ( $\bar{4}$ ).

## Describing symmetry operations with matrices



(for  $r = z$ )

# Rotations compatible with a lattice

Assume two lattice points, A and B, and that the minimum lattice spacing is **a** (unit translation). B generates a new point A' which is rotated from A by a generic angle  $\alpha$ . Applying the same rotational B operation R at A' generates a new point B'. If A' and B' are both lattice points then R is a symmetry operation. Due to the (translational) periodicity of the crystal, the new vector **ha,** which connects B and B', must be an integral multiple of a

```
ha 
          a 
  a \thetaA^{\bullet} \qquad a A^{\prime}B \leftarrow \mathbb{R}α α
   x \rightarrow 1θ θ
```

$$
\cos(\theta + \tfrac{\pi}{2}) = -\sin\theta
$$

```
AA' = aBB' = ha = a + 2xx = a.\sin(\theta) = -a.\cos(\theta + \pi/2) = -a.\cos(\alpha)ha = a - 2a \cdot cos(\alpha)ha - a = - 2a.cos(\alpha)
(h-1)/2 = -\cos\alphaFor h integer: h = -1, 0, 1, 2, 3Hence: cos\alpha = 0, \pm\frac{1}{2}, \pm 1\alpha = 0°, 60°, 90° 120°, 180°, or 360°
                Since an n-fold is 360%, this constrains a lattice
                to containing 1-, 2-, 3-, 4-, or 6-fold symmetry
```

```
(not 5 or 7, etc.)
```
# Rotations compatible with a lattice

Only 2, 3, 4 and 6-fold rotations can produce space filling patterns



# Point symmetry operations compatible with a lattice

### Rotations

2

3

4

6

- $\cdot$  1-fold  $360^\circ$ Identity I
- $\cdot$  2-fold  $180^\circ$
- $\cdot$  3-fold  $120^\circ$
- $\cdot$  4-fold  $90^\circ$
- $\cdot$  6-fold  $60^\circ$

## **Roto-Inversions** (Improper Rotations)

- $\overline{1}$  $\cdot$  1-fold  $360^\circ$  $\circ$
- $\overline{2}$  $\cdot$  2-fold  $180^\circ$
- $\overline{3}$  $\cdot$  3-fold  $120^{\circ}$  $\Delta$
- $\overline{4}$  $\cdot$  4-fold 90° ◈
- $\cdot$  6-fold  $\overline{6}$  $60^{\circ}$  $\bigcirc$

# Crystal systems

Crystals are axiomatically divided in 7 systems according to their symmetry



NB: Axiomatically = self-evident

# Symmetry operations compatible with the triclinic system

## other than 1 or 1 Only translational symmetry, no rotational symmetry

Illustrative 2D example (a planar lattice…)



An array of repeating motifs: neither the motif nor the lattice contains any elements of symmetry other than 1 or 1

# Symmetry operations compatible with the cubic system



# Crystal systems



# Centering

• What happens when other points are added to each of the previous lattices while maintaining the rotation symmetry (added at centered positions, centering involves only translation operations  $=$  centering operators)

• In each situation is it still a lattice? Is it a new lattice?

### Four possible lattice centerings

P: Primitive - lattice points on cell corners

I: Body-centred - additional lattice point at cell centre

F: Face-centred - one additional lattice point at centre of each face

A/B/C: Centred on a single face - one additional lattice point centred on A, B or C face



points within the unit cell is described by a set of **centering operators**: 

The location of the additional lattice

• Body centered (I) has additional lattice point at  $(1/2,1/2,1/2)$ 



• Face centered (F) has additional lattice points at  $(0,1/2,1/2)$ ,  $(1/2,0,1/2)$ , and  $(1/2,1/2,0)$ 



24 • Side centered (C) has an additional lattice point at  $(1/2,1/2,0)$ 

# **Centering**

Not all centering possibilities occur for each of the seven crystal systems: Only 14 unique combinations (Bravais lattices):

- Some centering types are not allowed because they would lower the symmetry of the unit cell (e.g. side centered cubic is not possible as this would destroy the three-fold symmetry that is an essential component of cubic symmetry)
- Some centering types are redundant (e.g. C-centered tetragonal can always be described using a smaller primitive tetragonal cell, see figure)



# Bravais lattices

The combination of crystal system and centering gives 14 Bravais lattices



A Bravais lattice is an infinite array of discrete points with **identical environment**: seven crystal systems + four lattice centering types = 14 Bravais lattices

# Point symmetry groups

A set of symmetry operations that leave an object invariant. Generically, there are infinite point symmetry groups. However, not all can be combined with a lattice.

In crystallography we are interested in objects that can be combined with the lattices: there are only 32 point groups compatible with periodicity in 3-D.

# Crystallographic point symmetry groups

- A crystallographic point group is a **set of symmetry operations**, like rotations or reflections, that leave a central point fixed while moving other directions and faces of the crystal to the positions of features of the same kind.
- For a true crystal the group must also be consistent with maintenance of the three-dimensional translational symmetry that defines crystallinity.
- The macroscopic properties of a crystal would look exactly the same before and after any of the operations in its point group. In the classification of crystals, each point group is also known as a crystal class.
- There are infinitely many three-dimensional point groups; However, the crystallographic restriction of the infinite families of general point groups results in there being only **32 crystallographic point groups.**

## The 32 point groups in stereographic projection





## Point Groups in Stereographic projection



# Point Groups in Stereographic projection





## Point Groups in Stereographic projection Tetragonal System











# Point Groups in Stereographic projection Hexagonal System  $a_{2}$  $a_2$  $a_{\hat{1}}$  $a_1$  $\overline{m}$











# In short...



### 32 point groups

# Space groups

Periodic solids have:

- lattice symmetry (purely translational)
- point symmetry (no translational component)
- possibly glide and/or screw axes (partly translational)

Together all the symmtery operations make up the space group 

# Glide planes

Combined reflections and translations (the translation is not a pure translational symmetry vector):



44 A stylised aerial view of a well coached 'eight', showing a translational symmetry operation: each rower is related to the next by a combination of translation and reflection.

# Glide planes



A glide plane. Translation from left to right across the page is accompanied by reflection through the plane of the paper.

# **Glide operations**

- · a-glide
	- Translate by  $\frac{1}{2}a$
- $\cdot$  b-glide
	- Translate by  $\frac{1}{2}$ b
- $\cdot$  c-glide
	- Translate by  $\frac{1}{2}c$
- $\cdot$  n-glide ( $\perp$  to a)
	- Translate by  $\frac{1}{2}$ b+  $\frac{1}{2}$ c
- n-glide ( $\perp$  to b)
	- Translate by  $\frac{1}{2}a + \frac{1}{2}c$
- n-glide ( $\perp$  to c)
	- Translate by  $\frac{1}{2}a + \frac{1}{2}b$
- · d-glide
	- Translate by  $\frac{1}{4}a+\frac{1}{4}b+\frac{1}{4}c$
- The glide translation must always be parallel to the glide plane.

## Screw axes

Combined rotations and translations (the translation is not a pure translational symmetry vector). The general symbol for a screw axis is  $N_n$ , where N is the order (2, 3, 4 or 6) of the axis, and *n* /*N* the translation distance expressed as a fraction of the repeat unit.

(g) - 0  $2<sub>1</sub>$  is a 180 $<sup>o</sup>$  rotation</sup> plus 1/2 cell translation (b)

47 (a) A two-fold screw axis,  $2<sub>1</sub>$ , shown perpendicular to the plane of the paper (left) and in the plane of the paper (right). Each half revolution is accompanied by a translation through half the repeat distance. (b) A fourfold screw axis,  $4<sub>1</sub>$ . (c) contracts and contracts

# Limitations on combination of symmetry elements

• Not all symmetry elements can be combined in the crystallographic point groups (only 32 point groups are compatible with periodicity in 3-D)

• Furthermore not all of the 32 point groups can be combined will all the lattices. For 3-D lattices there are:

- 14 Bravais lattices
- 32 point groups
- but only **230** space groups



### TABLE 3.4 Space Groups in Standard Orientations<sup>a</sup>

<sup>a</sup>The 11 Laue symmetries are separated by horizontal lines.

- All space group symbols start with a letter corresponding to the lattice centering, followed by a collection of symbols for symmetry operations in the three lattice directions.
- There are sometimes short notations for space groups symbols:
	- P*121* is usually written as P*2*
		- primitive cell
		- two-fold rotation along the b axis
	- $P2<sub>1</sub>2<sub>1</sub>2<sub>1</sub>$  (cannot be abbreviated)
		- primitive cell
		- $-2<sub>1</sub>$  screw along each axis, orthorhombic
	- C*mma* (full symbol: C*2/m2/m2/a*)
		- C-centered cell
		- mirror plane perpendicular to a
		- mirror plane perpendicular to b
		- glide plane perpendicular to c
		- other implied symmetry elements (e.g. 2-fold rotations)
	- P*nma*
		- primitive cell
		- n glide plane perpendicular to a
		- mirror plane perpendicular to b
		- glide plane perpendicular to c
		- other implied elements

### **Lattice centering**

- • **Primitive (P)**
- Base centering (C ٠
	- $x, y, z$
	- $x + \frac{1}{2}$ , y+ $\frac{1}{2}$ , z
- Body centering  $(I)$ 
	- $x, y, z$
	- $x + \frac{1}{2}$ ,  $y + \frac{1}{2}$ ,  $z + \frac{1}{2}$
- Face centering  $(F)$ 
	- $x, y, z$
	- $x + \frac{1}{2}$ ,  $y + \frac{1}{2}$ , z
	- $x + \frac{1}{2}$ , y, z+ $\frac{1}{2}$
	- $x, y + \frac{1}{2}, z + \frac{1}{2}$
- · Rhombohedral centering (R)
	- $x, y, z$
	- $x+1/3, y+2/3, z+1/3$
	- $x+2/2, y+1/2, z+2/2$

# Point Groups (Crystal Classes)

- Hermann-Mauguin Symbols (three positions)  $-$  Triclinic & monoclinic systems: one position
	- $-$  / means 'perpendicular to' as  $2/m$
	- $-$  Orthorhombic: three positions for a, b, c
	- Trigonal, Hexagonal, Tetragonal: 3 positions: c,  $a, [110]$
	- Cubic: 3 positions:  $[100]$  (a),  $[111]$ ,  $[110]$

Symmetry	Symbol	Designation If Parallel to Plane of Projection	Designation If Perpendicular to Plane of Projection
Center			
2-Fold axis			
3-Fold axis			
4-Fold axis			
6-Fold axis	6		
2-Fold screw axis	2 <sub>1</sub>		
3-Fold screw axis	3 <sub>1</sub>		
3-Fold screw axis	3 <sub>2</sub>		
4-Fold screw axis	4 <sub>1</sub>		
4-Fold screw axis	4 <sub>2</sub>		
4-Fold screw axis	$4_{3}$		≖
6-Fold screw axis	6		应
6-Fold screw axis	6 <sub>2</sub>		▲
6-Fold screw axis	63		
6-Fold screw axis	64		
6-Fold screw axis	6 <sub>5</sub>		ж
Mirror	$\boldsymbol{m}$		
a Glide plane	a		
<b><i>b</i></b> Glide plane	b		
$c$ Glide plane	с		
n Glide plane	n	N	
d Glide plane	d		$\rightarrow$ $\rightarrow$ $\rightarrow$

**TABLE 3.3 Symbols for Symmetry Elements** 

		<b>Equivalent Positions</b>	
Axis 2	Parallel to a	$x, y, z \quad x, \overline{y}, \overline{z}$	
$\overline{2}$	b	$x, y, z \overline{x}, y, \overline{z}$	
$\overline{2}$	$\boldsymbol{c}$	$x, y, z \overline{x}, \overline{y}, z$	
2 <sub>1</sub>	$\boldsymbol{a}$	$x, y, z \quad x + \frac{1}{2}, \bar{y}, \bar{z}$	
2 <sub>1</sub>	$\boldsymbol{b}$	$x, y, z \quad \bar{x}, y + \frac{1}{2}, \bar{z}$	
2 <sub>1</sub>	$\mathcal C$	$x, y, z \quad \bar{x}, \bar{y}, z + \frac{1}{2}$	
Plane m	Perpendicular to a	$x, y, z \overline{x}, y, z$	
m	$\boldsymbol{b}$	$x, y, z \quad x, \bar{y}, z$	
m	$\mathcal C$	$x, y, z \quad x, y, \overline{z}$	
$\boldsymbol{a}$	$\boldsymbol{b}$	$x, y, z \quad x + \frac{1}{2}, \bar{y}, z$	
$\boldsymbol{a}$	$\mathcal C$	$x, y, z \quad x + \frac{1}{2}, y, \bar{z}$	
$\boldsymbol{b}$	$\boldsymbol{a}$	$x, y, z \quad \bar{x}, y + \frac{1}{2}, z$	
$\boldsymbol{b}$	$\mathcal C$	$x, y, z \ x, y + \frac{1}{2}, \bar{z}$	
$\pmb{C}$	$\boldsymbol{a}$	$x, y, z \quad \bar{x}, y, z + \frac{1}{2}$	
$\mathcal C$	$\boldsymbol{b}$	$x, y, z$ , $x, \bar{y}, z + \frac{1}{2}$	
$\boldsymbol{n}$	$\boldsymbol{a}$	$x, y, z \quad \bar{x}, y + \frac{1}{2}, z + \frac{1}{2}$	
$\boldsymbol{n}$	$\boldsymbol{b}$	$x, y, z \quad x + \frac{1}{2}, \bar{y}, z + \frac{1}{2}$	
$\boldsymbol{n}$	$\mathcal{C}_{0}$	x, y, z $x + \frac{1}{2}$ , y + $\frac{1}{2}$ , $\bar{z}$	
$\boldsymbol{d}$	$\boldsymbol{a}$	$x, y, z \quad \bar{x}, y + \frac{1}{4}, z + \frac{1}{4}$	
$\boldsymbol{d}$	$\boldsymbol{b}$	$x, y, z \quad x + \frac{1}{4}, \bar{y}, z + \frac{1}{4}$	
$\boldsymbol{d}$	$\mathcal C$	$x, y, z \quad x + \frac{1}{4}, y + \frac{1}{4}, \overline{z}$	

**TABLE 3.2 Some Symmetry Elements and Their Equivalent Positions** 

53



P1, equivalent positions: (1) x, y, z

x, y, z are fractions of the length along each unit cell edge (values ranging from  $0.0$  to  $1.0$ )



P2<sub>1</sub>, equivalent positions:  $(1)$  x, y, z;  $(2)$  -x,  $y+1/2$ , -z

## International Tables for Crystallography - Volume A Entry for Space Group  $P2_1/c$  (#14)

Page 2

### Page 1



#### **Symmetry operations**

 $(1)$  1  $(2)$  2 $(0, \frac{1}{2}, 0)$  0, y,  $\frac{1}{4}$  $(3)$   $\bar{1}$  0.0.0 (4)  $c \, x, \frac{1}{4}, z$ 



Symmetry operations

 $(1)$   $(1)$  $(2) 20.10 0.1$  $0100$  $(0)$   $c$   $x_{12}$ 

57

 $\overline{a}$ 



# Wyckoff positions

- A useful piece of information contained in the International Tables are the Wyckoff positions that tell us where the atoms in a crystal can be found.
- The **letter** is simply a label and has no physical meaning. They are assigned alphabetically from the bottom up.
- The **multiplicity** tells us how many atoms are generated by symmetry if we place a single atom at that position.
- The **symmetry** tells us what symmetry elements the atom resides upon. The **uppermost** Wyckoff position, corresponding to an atom at an arbitrary **position never resides upon any symmetry elements**. This Wyckoff position is called the general position. The coordinates column tells us the coordinates of all of the symmetry related atoms
- All of the remaining Wyckoff positions are called **special positions**. They correspond to atoms which lie upon one of more symmetry elements, because of this they always have a smaller multiplicity than the general position. Furthermore, one or more of their fractional coordinates must be fixed otherwise the atom would no longer lie on the symmetry element.  $\frac{59}{2}$

# Asymmetric Unit

- Definition: smallest part of the unit cell which will generate the whole cell if all symmetry operators of the space groups are applied to it.
- Knowing the asymmetric unit and the symmetry of the structure allows generating the unit cell.



**Positions** 

Multiplicity, Wyckoff letter, Site symmetry

### **Systematic Absences** Reflection conditions Coordinates General:

**Reflection Conditions/** 



### Maximal isomorphic subgroups of lowest index

**Hc** [3]P 12<sub>1</sub>/c 1(b'=3b)(P2<sub>1</sub>/c); [2]P 12<sub>1</sub>/c 1(a'=2a or a'=2a, c'=2a+c)(P2<sub>1</sub>/c)

### Minimal non-isomorphic supergroups

- $[2]Pnna; [2]Pmna; [2]Pcca; [2]Pbam; [2]Pccn; [2]Pbcm; [2]Pnnm; [2]Pbcn; [2]Pbca; [2]Pnma;$ 1  $[2]Cmca$
- $[2] C 1 2/c 1 (C 2/c); [2] A 1 2/m 1 (C 2/m); [2] I 1 2/c 1 (C 2/c); [2] P 1 2_l/m 1 (2c' = c) (P 2_l/m);$ п  $[2]P12/c1(2b' = b)(P2/c)$

 $61$ 

if

 $Pn m a$ 

 $D_{2h}^{16}$ 

# Orthorhombic

No. 62











# Tetragonal





# Hexagonal



Directions and conventions for naming space group are the same as for trigonal.

**Cubic** 



# Describing crystals structures

- Full symmetry of a crystal is described by its space group
- The location of all atoms in a crystalline solid can be specified by a combination of all the symmetry elements and the fractional coordinates for a unique set of atoms (asymmetric unit)

We specify the atomic coordinates for a small number of atoms. Then we apply all the symmetry elements including the lattice symmetry to build up the full 3D structure. 

N.B.: Each lattice point may be associated with many atoms





# Asymmetric units...

# Examples

**Ta adopts the Ta-type structure with space group Im3m (229) with atoms at 2a (0,0,0) and a=0.33 nm.**

Ti adopts the Mg-type structure with space group p6<sub>3</sub>/mmc **(194) with atoms at 2c (1/3,2/3,1/4) and a=0.295 nm and c=0,4686 nm.**

**Si adopts the diamond-type structure with space group Fd3m (227) with atoms at (16c) 1/8,1/8,1/8 and a=0.543 nm.**

**FeO adopts the NaCl-type structure with O in Cl sites (only lattice parameter missing…).**